

Structural evolution and metallicity of lead clusters

Götz, Daniel A; Shayeghi, Armin; Johnston, Roy L; Schwerdtfeger, Peter; Schäfer, Rolf

DOI:

[10.1039/c6nr02080a](https://doi.org/10.1039/c6nr02080a)

License:

Creative Commons: Attribution (CC BY)

Document Version

Publisher's PDF, also known as Version of record

Citation for published version (Harvard):

Götz, DA, Shayeghi, A, Johnston, RL, Schwerdtfeger, P & Schäfer, R 2016, 'Structural evolution and metallicity of lead clusters', *Nanoscale*, vol. 8, no. 21, pp. 11153-60. <https://doi.org/10.1039/c6nr02080a>

[Link to publication on Research at Birmingham portal](#)

General rights

Unless a licence is specified above, all rights (including copyright and moral rights) in this document are retained by the authors and/or the copyright holders. The express permission of the copyright holder must be obtained for any use of this material other than for purposes permitted by law.

- Users may freely distribute the URL that is used to identify this publication.
- Users may download and/or print one copy of the publication from the University of Birmingham research portal for the purpose of private study or non-commercial research.
- User may use extracts from the document in line with the concept of 'fair dealing' under the Copyright, Designs and Patents Act 1988 (?)
- Users may not further distribute the material nor use it for the purposes of commercial gain.

Where a licence is displayed above, please note the terms and conditions of the licence govern your use of this document.

When citing, please reference the published version.

Take down policy

While the University of Birmingham exercises care and attention in making items available there are rare occasions when an item has been uploaded in error or has been deemed to be commercially or otherwise sensitive.

If you believe that this is the case for this document, please contact UBIRA@lists.bham.ac.uk providing details and we will remove access to the work immediately and investigate.

Cite this: *Nanoscale*, 2016, **8**, 11153

Structural evolution and metallicity of lead clusters†

Daniel A. Götz,^{*a} Armin Shayeghi,^a Roy L. Johnston,^b Peter Schwerdtfeger^c and Rolf Schäfer^a

The evolution of the metallic state in lead clusters and its structural implications are subject to ongoing discussions. Here we present molecular beam electric deflection studies of neutral Pb_N ($N = 19\text{--}25, 31, 36, 54$) clusters. Many of them exhibit dipole moments or anomalies of the polarizability indicating a non-metallic state. In order to resolve their structures, the configurational space is searched using the Pool Birmingham Cluster Genetic algorithm based on density functional theory. Spin-orbit effects on the geometries and dipole moments are taken into account by further relaxing them with two-component density functional theory. Geometries and dielectric properties from quantum chemical calculations are then used to simulate beam deflection profiles. Structures are assigned by the comparison of measured and simulated beam profiles. Energy gaps are calculated using time-dependent density functional theory. They are compared to Kubo gaps, which are an indicator of the metallicity in finite particles. Both, experimental and theoretical data suggest that lead clusters are not metallic up to at least 36 atoms.

Received 11th March 2016,

Accepted 28th April 2016

DOI: 10.1039/c6nr02080a

www.rsc.org/nanoscale

1. Introduction

As clusters grow towards the bulk, their geometric and electronic structures undergo fundamental transitions. In particular, the evolution of the metallic state in metal atom clusters has drawn attention. The discussion of metallicity on the nanoscale is hampered by the fact that it is well defined only for the bulk, *i.e.* by a non-zero electronic density of states at the Fermi energy. Generally, clusters have a gap at E_F because of their finite size and so they are never metallic in a strict sense. Issendorff and Cheshnovsky circumvented this issue by considering a cluster as metallic if its energy level spacing E_{gap} at the Fermi energy drops below the Kubo gap.¹ The Kubo gap represents the average energy level spacing in a metallic particle of finite size, if a statistical level distribution is assumed.² However, the electronic structure of simple metal clusters like sodium is well described within the shell model where electrons are treated as free particles in a spherical box. This

implicates highly degenerate energy levels and in contrast to the Kubo criteria substantial gaps for certain cluster sizes arise. Considering the Kubo criteria solely is therefore not sufficient to assess the metallic character of a cluster. This point has been discussed for zinc clusters thoroughly.³ Besides the electronic shell structure and the Kubo gap also the electronic screening and the bonding nature have to be taken into account. Although the concept of metallicity has to be used cautiously, we term a cluster metallic in the following if the mentioned criteria are sufficiently fulfilled.

Experimental values for E_{gap} can be derived from photoelectron spectroscopy (PES). The difference of the first two vertical photodetachment energies is the gap of the corresponding neutral cluster in the geometry of the anion. This gap is a reasonable approximation to the gap of the neutral cluster, if the ground state geometries of the anion and the neutral cluster do not differ substantially. Characteristic metal to insulator transitions (MIT) have been detected by PES for several cluster species. In Hg_N^- clusters for example, the gap steadily decreases with the cluster size and closes at about $N = 400$.⁴ The gaps of Sn_N^- clusters on the other hand close at around $N = 40$.⁵ In the same size regime the geometries of Sn_N^+ clusters become spherical whereas prolate structures are preferred for smaller species as ion mobility measurements have shown.⁶ Although the charge state certainly influences the geometric structure of a cluster the relation between geometric structure and metallicity becomes evident here. The structure and properties of the next heavier homolog Pb are strongly influenced by spin-orbit (SO) effects.⁷ It is still not

^aEduard-Zintl-Institut, Technische Universität Darmstadt, Alarich-Weiss-Straße 8, 64287 Darmstadt, Germany. E-mail: goetz@cluster.pc.chemie.tu-darmstadt.de; Fax: +49 6151 1623841; Tel: +49 6151 1623845

^bSchool of Chemistry, University of Birmingham, Edgbaston, Birmingham B15 2TT, UK

^cCentre for Theoretical Chemistry and Physics, The New Zealand Institute for Advanced Study, Massey University (Albany), Bob Tindall Bldg., 0745 Auckland, New Zealand

† Electronic supplementary information (ESI) available: Beam profile simulations for rotational temperatures $T_{\text{rot}} = 5\text{--}50$ K, the beam profile of Pb_{54} and cluster coordinates in Å. See DOI: 10.1039/C6NR02080A

clear how the metallic state evolves in Pb clusters, *i.e.* if the MIT occurs at a certain critical size or by gradual gap closure. Early PES studies did not come to a clear conclusion.^{8,9} Later Senz *et al.* found reduced core-hole screening by PES as the cluster size drops below 20 atoms, indicating a MIT in this size regime.¹⁰ Recently, Heinzelmann *et al.* presented a time-dependent PES study where relaxation life times denote a non-metallic electronic structure for Pb₂₈[−].¹¹ Despite the comprehensive experimental work on Pb_N[−] clusters it remains unclear where the MIT occurs.

In the present article we study neutral Pb_N clusters, which is straightforward since any influence of the excess charge is avoided. Several theoretical studies are dedicated to neutral Pb cluster.^{12–14} Wang *et al.* calculated gaps between highest occupied (HOMO) and lowest unoccupied molecular orbitals (LUMO) as well as densities of states up to Pb₂₂ using density functional theory (DFT).¹⁵ They observed gaps around 1 eV for their largest cluster investigated and a considerable dependence on the geometric structure. However, the reliability of HOMO–LUMO gaps from DFT calculations has been questioned¹⁶ and they should be treated with caution.

In this study we employ the electric molecular beam deflection (MBD) method in order to shed light on the MIT in Pb clusters and its structural manifestation. The MBD probes dipole moments and polarizabilities of a given neutral cluster. Since a classical metallic particle cannot have a permanent dipole moment, *i.e.* the electronic screening should be almost perfect, it is a meaningful indicator for a non-metallic state. This was concisely pointed out by de Heer and co-workers for small sodium clusters.¹⁷ Beam profiles from MBD experiments are also sensitive for the cluster geometry. Hence, a structural assignment is possible, as demonstrated for several group 14 clusters.^{18–21} Former molecular beam deflection experiments on neutral lead clusters at 50 K indicated dipole moments or anomalies in the polarizability for several Pb_N (*N* ≤ 40) without suggesting any structures for them.²² We have repeated these measurements at 30 K to ensure the rigidity of the clusters. The experimental data are supported by an extensive global minimum search. Structures are further relaxed employing two-component (2c) DFT in order to account for SO effects on the geometries and dipole moments. The metallic character of the clusters is further evaluated by gaps from 2c time-dependent DFT (2c-TDDFT) calculations.

2. Experimental and computational methods

The experimental²³ and computational²¹ procedures have been described earlier and we only give a brief outline here. Lead clusters are generated in a pulsed laser vaporization source using helium as carrier gas.²⁴ The cluster–helium mixture is cooled in a cryogenic nozzle (*T*_{nozzle} = 30 K) before a molecular beam is formed by supersonic expansion into a high vacuum chamber. Two collimators shape the molecular beam to a rectangular profile before it enters an electric two-wire field unit.

The external field with a gradient $\frac{\partial E}{\partial z}$ deflects a cluster in a quantum state *i* with mass *m* and velocity *v* by

$$d_i = -\frac{A}{mv^2} \frac{\partial E}{\partial z} \frac{\partial \epsilon_i}{\partial E}. \quad (1)$$

Here, *A* is an apparatus constant and $\frac{\partial \epsilon_i}{\partial E}$ the Stark effect. The latter is a cluster specific quantity and determines its deflection behaviour. Downstream of the deflection unit the clusters arrive at a scanning slit plate, are photoionized by a F₂ excimer laser (7.89 eV) and subsequently detected by a time-of-flight mass spectrometer. Photoionisation mass spectra are measured as a function of the slit position with and without applied electric field. The relative intensities of a cluster species in this mass spectra yield beam profiles as shown in Fig. 1a and 2a. The polarizability of a rigid, almost spherical rotor can be extracted from the beam profile by first order perturbation theory.²² Herein, experimental polarizabilities α_{exp} are reported for Pb_N (*N* = 20–24), where no or a minor dipolar contribution to the beam shift due to a permanent dipole moment is observed.²⁵

Promising lead cluster structures are obtained by a global optimization approach employing the Pool Birmingham Cluster Genetic Algorithm.²⁶ Structures are relaxed using plane-wave DFT within in the Quantum Espresso program package.²⁷ The Perdew–Burke–Ernzerhof²⁸ (PBE) functional is used and 14 electrons for each lead atom are treated explicitly; the remaining core electrons are described by a pseudopotential^{29–31} with a suggested minimum cutoff of 40 Ry of the plane-wave basis. The following orbital based DFT calculations (PBE) are carried out using the Turbomole³² and NWChem³³ program packages. Isomers within 0.5 eV of the global minimum (GM) are considered for further local optimizations at the scalar relativistic DFT level of theory with the def2-TZVP basis set³⁴ and the Stuttgart energy consistent effective core potential (ECP).³⁵ Vibrational analyses are performed to ensure all geometries correspond to true minima. These structures are used as starting geometries for 2c-DFT local optimizations (def2-TZVP-2c/Stuttgart SO-ECP).^{35–37} The results are shown in Fig. 1b and 2b (see the ESI† for the corresponding xyz coordinates).

Static isotropic polarizabilities α_{el} are calculated for the 2c minimum structures at the scalar relativistic level of theory only. In order to estimate the gaps of the lead clusters 2c-TDDFT calculations have been carried out.³⁸ Gaps from TDDFT calculations are expected to yield more accurate results than just taking HOMO–LUMO gaps from DFT calculations.¹⁶

The deflection of an ensemble of rigid rotors in an electric field can be simulated by a convolution of the undeflected profile with the dipole distribution function. The latter is readily obtained by a molecular dynamics simulation, where moments of inertia, dipole moments and polarizabilities from DFT calculations enter parametrically.¹⁹ The only free parameter in this simulation is the rotational temperature of the clusters. So, simulations are carried out for each isomer from *T*_{rot} = 5–50 K. However, in accordance to previous studies *T*_{rot} = 30 K is appropriate (see Fig. S1†).



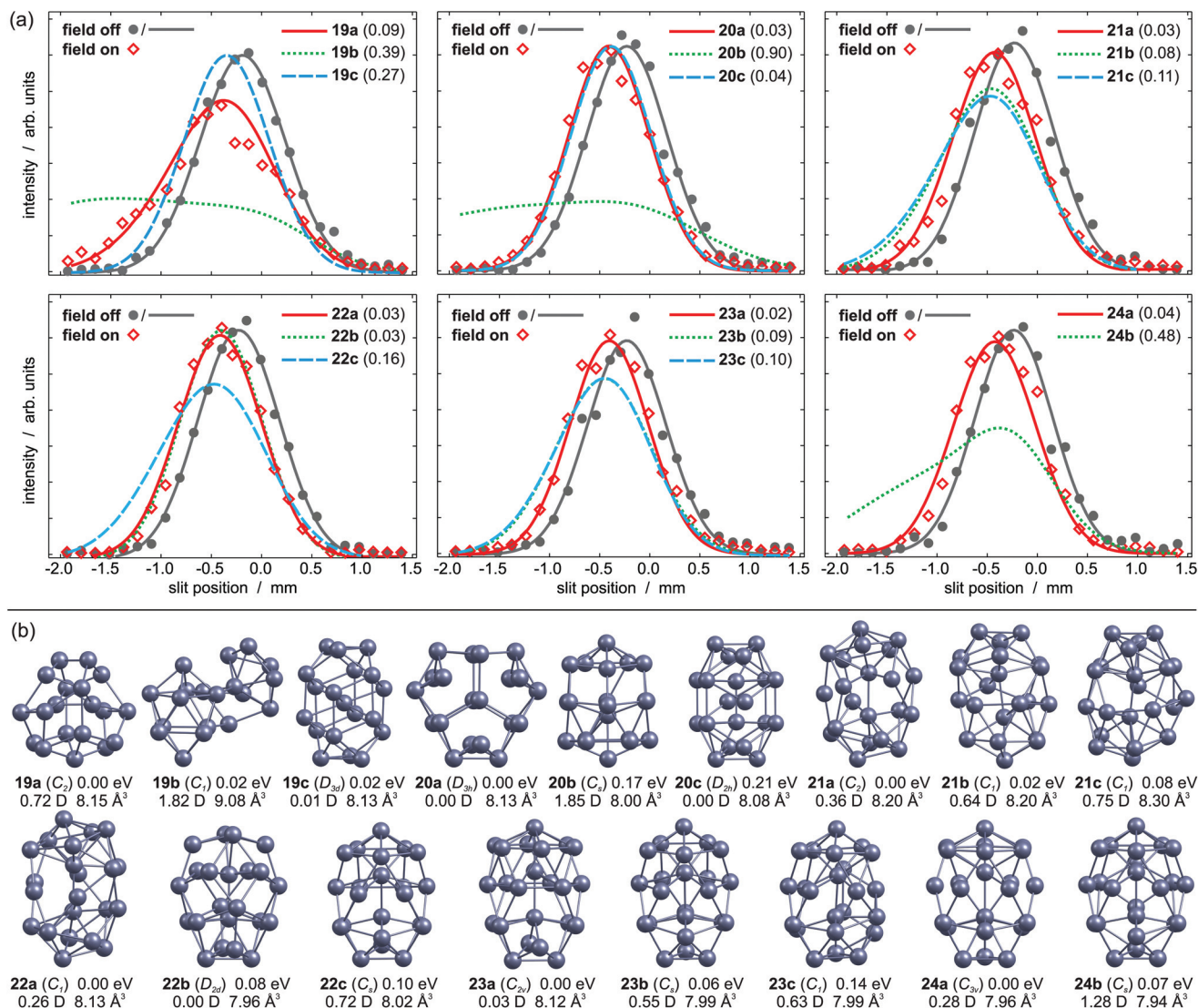


Fig. 1 (a) Experimental (grey circles: field off, red diamonds: field on) and simulated beam profiles of Pb₁₉–Pb₂₄ (coloured curves) in dependence on the slit position. A gaussian is fitted to the field off values as guide to the eye (grey curve). Numbers in brackets give the least square fit of the simulated profiles to the experimental data points. (b) Local minimum structures of Pb₁₉–Pb₂₄, including their relative energies, dipole moments and polarizabilities.

3. Results and discussion

3.1. Structures and dielectric properties

The experimental beam profiles of Pb₁₉ are shown in Fig. 1a. A broadening is observed as the field is applied. This is characteristic for a cluster with permanent dipole moment. Fig. 1b displays the three lowest lying isomers identified by the GA run. The GM **19a** is a C₂ symmetric structure similar to the GM reported by Ho and coworkers.¹³ It has a moderate dipole moment of 0.72 D and its simulated beam profile well describes the experimental data. Isomer **19b** is a peculiar structure consisting of Pb₉ and Pb₁₀ subunits only 0.02 eV above the GM; this geometry has a dipole moment of 1.82 D and the corresponding beam profile is significantly broadened. Therefore, it can be ruled out. Almost equal in energy is

isomer **19c**, an atom-centered prolate D_{3d} structure with a vanishing dipole moment. This isomer does not fit the experimental data and so **19a** is expected to be the dominant structure in the molecular beam.

The three energetically lowest lying isomers of Pb₂₀ are shown in Fig. 1b. The GM structure **20a** is an oblate D_{3h} symmetric cluster with a zero dipole moment. This is in very good agreement with the non-broadened experimental beam profile (Fig. 1a, $\alpha_{\text{exp}} = 7.4 \pm 1.4 \text{ \AA}^3$). Isomer **20b** is 0.17 eV higher in energy and contains a hexagonal antiprism motif. It has a considerable dipole moment of 1.85 D, hence a substantial contribution to the experimental beam profile is excluded. Isomer **20c** is a prolate structure with two central atoms and has previously been proposed as the GM.³⁹ Although it nicely matches the experimental beam profile, it seems unlikely to be the

dominating structure under our experimental conditions due to its high relative energy.

The beam profile of Pb_{21} in Fig. 1a is slightly broadened if the deflecting field is on, pointing towards a structure with small dipole moment ($\alpha_{\text{exp}} = 7.2 \pm 1.3 \text{ \AA}^3$). The C_2 symmetric GM **21a** is a prolate structure with two inner atoms and can be considered as two isolated pentagonal bipyramids (PBP) with seven glue atoms. It has a small dipole moment of 0.36 D and its simulated beam profile explains the experimental one very well. The higher lying isomers **21b** and **21c** can be both ruled out due to the increased beam broadening in the simulation. Hence, **21a** appears to be the only relevant isomer in the experiment.

The GM of Pb_{22} is a C_1 symmetric structure **22a**, which is also shown in Fig. 1b. It resembles the motif of **21a** with two individual PBP plus eight glue atoms and the simulated beam profile is in very good agreement with the experiment. However, isomer **22b** has no permanent dipole moment due to its symmetry (D_{2d}). No remarkable broadening is observed for the experimental beam profile ($\alpha_{\text{exp}} = 6.9 \pm 0.9 \text{ \AA}^3$), which is very well reproduced by the simulated beam profile. Both, **22a** and **22b**, sufficiently explain the experimental data and cannot be distinguished in our experiment, though the small α_{exp}

probably hints towards **22b**. The higher lying isomer **22c** can be ruled out based on its large dipole moment.

The experimental beam profile of Pb_{23} is shifted but not broadened, so a structure with a small or zero dipole moment is expected ($\alpha_{\text{exp}} = 6.4 \pm 0.9 \text{ \AA}^3$). This applies for the GM **23a**, which contains a hexagonal antiprism motif. Its beam profile is in excellent agreement with the experiment. Isomer **23b** has a very similar structure to **22c**, but its dipole moment is too large to explain the experiment. Isomer **23c** is a distorted version of **23b** and cannot describe the experiment either.

No pronounced beam broadening is observed for Pb_{24} ($\alpha_{\text{exp}} = 6.7 \pm 1.1 \text{ \AA}^3$). Only two local minimum structures are found below 0.2 eV. The putative GM **24a** is a prolate C_{3v} symmetric structure, which is very similar to **23a** with respect to the hexagonal antiprism motif. Its small dipole moment of 0.28 D causes a slightly broadened beam profile, which nicely explains the experiment. The structure of isomer **24b** follows the pattern of **22c** and **23b**. Its considerable dipole moment of 1.28 D yields a significantly broadened beam profile in contrast to the experiment.

Fig. 2b displays energetically low lying local minimum structures and beam profiles of Pb_{31} . The GM is the C_1 symmetric structure **31a** containing three PBP subunits. Its

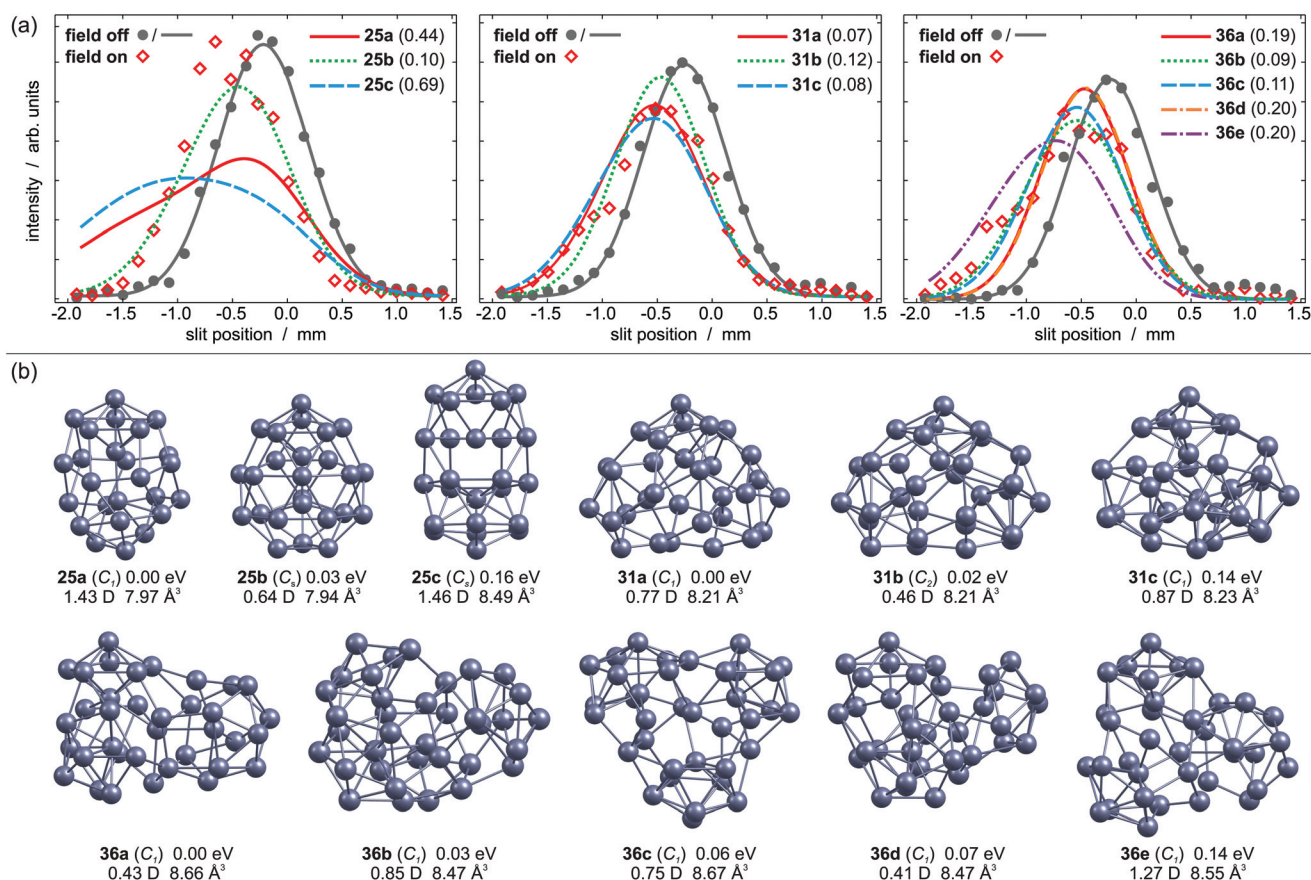


Fig. 2 (a) Experimental (grey circles: field off, red diamonds: field on) and simulated beam profiles of Pb_{25} , Pb_{31} and Pb_{36} (coloured curves) in dependence on the slit position. A gaussian is fitted to the field off values as guide to the eye (grey curve). Numbers in brackets give the least square fit of the simulated profiles to the experimental data points. (b) Local minimum structures of Pb_{25} , Pb_{31} and Pb_{36} , including their relative energies, dipole moments and polarizabilities.



simulated beam profile is in reasonable agreement with the experimentally observed profile, which is substantially broadened. Isomer **31b** is only 0.02 eV higher in energy. Its geometry is very similar to **31a**, but it has a smaller dipole moment. This gives rise to a less broadened beam profile than the one observed experimentally. Isomer **31c** has a similar dipole moment compared to **31a**, so it is also able to explain the experimental findings. Nevertheless, a significant contribution to the experimental beam profile seems unlikely due to its high relative energy.

The Pb_{36} cluster shows a distinctive beam broadening if the electric field is on (Fig. 2a). In Fig. 2b the five lowest lying local minimum structures are reported. Their geometries all have C_1 symmetry and permanent dipole moments ranging from 0.41 to 1.27 D. The GM **36a** ($\mu_0 = 0.43$ D) contains three independent PBP, but it can also be considered as three Pb_9 subunits with nine glue atoms. The Pb_9 cluster on the other hand consists of two interpenetrating PBP.²¹ However, the dipole moment of **36a** is too small to explain the experimental beam broadening. Isomer **36b** is only 0.03 eV higher in energy than **36a** and has a dipole moment of 0.85 D. Its simulated beam profile resembles the experimental data very well. The structure also contains three Pb_9 subunits. This applies also for isomer **36c**, though the different relative arrangement gives rise to a dipole moment of 0.75 D, which still yields a reasonable agreement with the experiment. Isomer **36d** fits the experimental data in similar quality as **36a** and can therefore be ruled out. The simulated beam profile of **36e** is too broad to explain the experiment. Isomer **36b** appears to be predominant in the molecular beam though **36c** cannot be ruled out.

The beam profiles of Pb_N ($N = 19\text{--}24, 31, 36$) are well described by rigid rotor simulations employing the structures and properties of low lying local minima. As shown in Fig. 2a this procedure does not hold for Pb_{25} . Its beam profile is strongly shifted indicating an elevated polarizability, but is not broadened at all. The three energetically lowest lying isomers exhibit dipole moments between 0.64–1.46 D. Although the permanent dipole moment of a rigid rotor contributes to the total shift, a considerable beam broadening would be expected as well.²⁵ Hence, none of them fits the measured beam profile within the rigid rotor simulation. The observed beam shift comes up to an effective polarizability of $10.90 \pm 1.16 \text{ \AA}^3$, whereas DFT predicts values of 7.9–8.5 Å^3 . Since the polarizability of a cluster usually does not depend strongly on the geometry, it seems very unlikely that an isomer without permanent dipole moment and an increased polarizability has been missed out by the global optimization. In fact, more than 20 local minimum structures up to 0.5 eV have been found, but none of them provides a good fit to the experiment.

The dipole moment of a rigid cluster is fixed with respect to its geometry and the projection of the dipole moment on the field axis only depends on the rotational motion. In a thermally excited cluster, vibrational motions can couple to the rotation and induce or alter the dipole moment components. If these processes are not correlated to the rotational motion of the cluster, the beam broadening completely vanishes and

only the beam shift is enhanced by a contribution of the permanent dipole moment μ_0 . The resulting effective polarizability α_{eff} is described in a Langevin–Debye-type model

$$\alpha_{\text{eff}} = \alpha_{\text{el}} + \frac{\mu_0^2}{3k_{\text{B}}T_{\text{int}}}. \quad (2)$$

The internal temperature T_{int} is assumed to be equal to the nozzle temperature (30 K). Taking a dipole moment of $\mu_0 = 1.43$ D into account results in an effective polarizability of 14.55 Å^3 for isomer **25a**. Isomer **25b**, which is only 0.03 eV higher in energy, has a dipole moment of 0.64 D, giving rise to an effective polarizability of 10.20 Å^3 . This value is in good agreement with the experiment, whereas α_{eff} of isomer **25c** is too large (15.35 Å^3). The Langevin–Debye-type behaviour has been described for the Rh_{10} cluster at 49 K (ref. 40) and for silicon clusters at room temperature.⁴¹ At room temperature multiple vibrational modes are excited and isomerization processes become possible. This is certainly not the case for Pb_{25} at 30 K so the question persists why no beam broadening is observed for this cluster. It has been shown for organic molecules that the beam broadening can be quenched if certain vibrational modes are excited, which couple strongly to the rotational motion.⁴² Fig. 3 displays harmonic frequency analyses of the relevant cluster species up to 50 cm^{-1} . While all clusters show at least one weak IR active normal mode below 30 cm^{-1} , only **25b** has three intense modes in this range. Two are very close in energy at about 26.7 (A'') and 26.9 cm^{-1} (A'), respectively. The third mode is very soft (2.6 cm^{-1}) and corresponds to a hindered rotation of the central dimer relative to the cage (A''). The excitations of these modes suffice to fully quench the beam broadening. Since all other clusters also have vibrational modes below 30 cm^{-1} we cannot exclude that the beam broadening is partially quenched for them, *i.e.* the experimental beam profile is less broadened than expected for a rigid cluster.

The results of the MBD experiments reveal that several Pb_N clusters ($N = 19, 25, 31, 36$) have significant dipole moments. Since an excellent electronic shielding, *i.e.* a vanishing dipole moment is expected for a metallic cluster, this strongly suggests that they are not metallic. The clusters Pb_N ($N = 20\text{--}24$) do not show a significant beam broadening or increased beam shift, what actually also applies for $N = 26\text{--}30$,

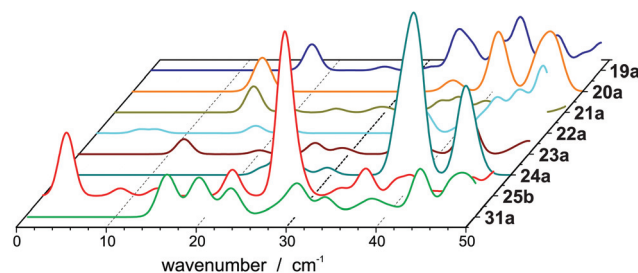


Fig. 3 Vibrational spectra of Pb_N ($N = 19\text{--}25, 31$) below 50 cm^{-1} .



32–34, which have been omitted in this study due to the excessive computational cost of the global optimization. Since a vanishing permanent dipole moment is not a sufficient condition of metallicity their character cannot be assessed based on the MBD experiments only. However, the elucidated geometries give further insight. Li *et al.* proposed that the electron delocalization is obstructed by individual, strong covalently bound PBP Pb_7 subunits.³⁹ The occurrence of such structural subunits in Pb clusters would further support that they are not metallic. Apart from **19a** and **20a** all experimentally identified isomers contain several PBP. In particular, Pb_N ($N = 21$ –25) can be considered as two PBP with a growing number of glue atoms. Isomer **31a** contains three PBP, though four are found in **31c**. Despite the fact that the PBP is the dominant structural subunit, their number is not necessarily maximised in the GM. The situation is consistent for Pb_{36} , though Pb_9 subunits appear, which are basically two interpenetrating PBP. We note that the occurrence of subunits with localised electron density is of course related to the appearance of permanent dipole moments, *i.e.* due to the relative arrangement of the subunits in a certain cluster. However, for some cluster species the permanent dipole moment (almost) vanishes for symmetry reasons although they contain PBP subunits.

3.2. TDDFT gaps

The experimentally observed dipole moments and the assigned structures show that lead clusters up to 36 atoms do not fulfill typical criteria of metallicity. Further evidence for their metallic character comes from the size of the energy gap E_{gap} compared to the Kubo gap E_{Kubo} , which is larger than E_{gap} in a metallic particle

$$E_{\text{Kubo}} = \frac{4E_{\text{F}}}{3N_{\text{E}}} \quad (3)$$

Gaps are calculated by 2c-TDDFT for Pb_N using experimentally proven geometries from section 1 and the literature.²¹ Fig. 4 displays the results together with the Kubo gaps, taking four valence electrons per lead atom (N_{E}) into account ($E_{\text{F}}(\text{Pb}) = 9.37$ eV). The gaps vary with the cluster size and decrease from 0.75 eV for Pb_7 to 0.26 eV for Pb_{36} with relative maxima for Pb_9 , Pb_{13} , Pb_{19} . In the jellium model maximum gaps are expected for Pb_N ($N = 9, 10, 17, 20, \dots$) due to shell closures. This is certainly not reflected by the data in Fig. 4. The gaps do not drop below the Kubo gap for any cluster size, which decreases from 0.45 for Pb_7 to 0.09 eV for Pb_{36} . The Pb_{23} has the smallest gap of all clusters considered (0.20 eV), which is only slightly larger than the corresponding Kubo gap (0.14 eV). Accordingly, the sizes of the gaps further corroborate the experimental results and show that also clusters can be considered as non-metallic, which do not have a significant permanent dipole moment. The correlation between the gaps and the dipole moments does not follow a clear trend. For the icosahedron Pb_{13} a zero dipole moment and the largest gap (0.92 eV) is observed, while other cluster like Pb_{12} ($\mu_0 = 0.59$ D) and Pb_{19} have large gaps and significant dipole moments. Nevertheless, we mention, that the trend of gaps and dipole

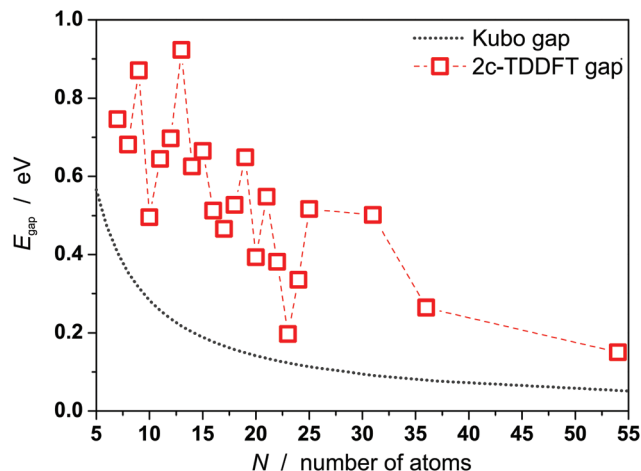


Fig. 4 Kubo gaps for Pb_N ($N = 7$ –25, 31, 36, 54) and gaps from 2c-TDDFT calculations. Apart from Pb_{54} the experimentally found geometries are used, which correspond to the GM or the GM + 1 for Pb_{16} , Pb_{18} and Pb_{25} , respectively. Structures of Pb_7 – Pb_{18} are taken from the literature.²¹ For Pb_{54} a cuboctahedral geometry is assumed.³⁷

moments is qualitatively similar for Pb_N ($N = 19$ –25), *i.e.* clusters with small dipole moments have also small gaps (*e.g.* Pb_{23}) and *vice versa*. This is interesting insofar as dipole moments and gaps vanish in the bulk limit.

In order to estimate the evolution of the gaps for even larger sizes, the gap of Pb_{54} has been calculated. A global optimization as outlined in section 2 was not feasible here, but Pb_{54} is assumed to be a cuboctahedron.³⁷ The dielectric properties of the local optimized geometry are $\mu_0 = 0.00$ D and $\alpha_{\text{el}} = 8.42 \text{ \AA}^3$ ($\alpha_{\text{exp}} = 7.6 \pm 0.7 \text{ \AA}^3$, see Fig. S2†). It has a gap of 0.15 eV, what demonstrates that 2c-TDDFT gaps further decrease with increasing cluster size. Further, it is already very close to the Kubo gap (0.06 eV) indicating the onset of metallicity in this size regime.

4. Conclusions and outlook

We reported beam profiles from electric molecular beam deflection at 30 K for several medium sized lead clusters. Cluster structures are assigned to the observed beam profiles by the comparison of experimental with simulated beam profiles. For Pb_{19} and Pb_{20} we find an asymmetric and an oblate structure, respectively as GM. The trend towards atom centered structures starting with Pb_{13} is interrupted for Pb_{19} and Pb_{20} . Prolate structures with two central atoms are identified for Pb_N ($N = 21$ –25). The Pb_{25} cluster is a special case here, since it is not deflected like a rigid rotor, *i.e.* the beam broadening is completely quenched. This Langevin–Debye-type deflection suggests that Pb_{25} is fluxional already at 30 K. The vibrational analysis shows that Pb_{25} has a prominent IR active mode below 30 cm^{-1} . Its excitation is obviously sufficient to suppress any beam broadening. This makes Pb_{25} an interesting starting point for further investigations of the deflection behaviour and the dynamics of flexible clusters.



The metallicity of lead clusters has been evaluated in terms of their electron screening, their gaps and their structures. The observed dipole moments indicate an electronic screening which is typical for non-metallic systems. Gaps from TDDFT calculations show that they are considerably larger than the Kubo gaps. We have identified Pb₇ PBP as predominating structural subunits, which have earlier been related to strong electron localization.³⁹ With respect to their experimentally observed electronic screening behaviour and their gaps in comparison to the Kubo gap we conclude that lead clusters are not metallic up to at least Pb₃₆. The non-metallicity of lead clusters is structurally not manifested in prolate geometries as in tin clusters but in the cumulation of Pb₇ or Pb₉ subunits. As quantum chemistry suggests the gaps further decrease for larger cluster sizes (Pb₅₄). Finally we notice that spin-orbit coupling leads to a separation of 0.97 eV between the state with total angular momentum quantum number $J = 0$ ($p_{1/2}^2$) and $J = 1$ ($p_{1/2}^1 p_{3/2}^1$) in the ³P ground state of atomic lead.⁴³ This renders Pb effectively as a closed-shell atom undergoing rather weak bonding in clusters and perhaps explains their increased fluxionality. This spin-orbit gap could substantially delay the onset of metallicity in lead clusters.

Acknowledgements

We acknowledge financial support by the DFG (grant SCHA 885/7-3 and SCHA 885/10-2). D. A. G. is grateful for a scholarship of the Cusanuswerk. The calculations reported here are performed on the following HPC facilities: The University of Birmingham BlueBEAR facility; the MidPlus Regional Centre of Excellence for Computational Science, Engineering and Mathematics, funded under EPSRC grant EP/K000128/1; and via our membership of the UK's HPC Materials Chemistry Consortium, which is funded by EPSRC (EP/L000202). This work made use of the facilities of ARCHER, the UK's national high-performance computing service, which is funded by the Office of Science and Technology through EPSRC's High End Computing Programme; and Simurg, the High-Performance Supercomputer of Massey University.

References

- 1 B. von Issendorff and O. Cheshnovsky, *Annu. Rev. Phys. Chem.*, 2005, **56**, 549–580.
- 2 R. Kubo, *J. Phys. Soc. Jpn.*, 1962, **17**, 975–986.
- 3 A. Aguado, A. Vega, A. Lebon and B. von Issendorff, *Angew. Chem., Int. Ed.*, 2015, **54**, 2111–2115.
- 4 R. Busani, M. Folkers and O. Cheshnovsky, *Phys. Rev. Lett.*, 1998, **81**, 3836–3839.
- 5 L.-F. Cui, L.-M. Wang and L.-S. Wang, *J. Chem. Phys.*, 2007, **126**, 064505.
- 6 A. A. Shvartsburg and M. F. Jarrold, *Phys. Rev. A*, 1999, **60**, 1235–1239.
- 7 A. Hermann, J. Furthmüller, H. W. Gäggeler and P. Schwerdtfeger, *Phys. Rev. B: Condens. Matter*, 2010, **82**, 155116.
- 8 G. Ganteför, M. Gausa, K.-H. Meiwes-Broer and H. O. Lutz, *Z. Phys. D: At., Mol. Clusters*, 1989, **12**, 405–409.
- 9 C. Lüder and K.-H. Meiwes-Broer, *Chem. Phys. Lett.*, 1998, **294**, 391–396.
- 10 V. Senz, T. Fischer, P. Oelßner, J. Tiggesbäumker, J. Stanzel, C. Bostedt, H. Thomas, M. Schöffler, L. Foucar, M. Martins, J. Neville, M. Neeb, T. Möller, W. Wurth, E. Rühl, R. Dörner, H. Schmidt-Böcking, W. Eberhardt, G. Ganteför, R. Treusch, P. Radcliffe and K.-H. Meiwes-Broer, *Phys. Rev. Lett.*, 2009, **102**, 138303.
- 11 J. Heinzelmann, P. Kruppa, S. Proch, Y. D. Kim and G. Ganteför, *Chem. Phys. Lett.*, 2014, **603**, 1–6.
- 12 J. P. K. Doye and S. C. Hendy, *Eur. Phys. J. D*, 2003, **22**, 99–107.
- 13 X.-P. Li, W.-C. Lu, Q.-J. Zang, G.-J. Chen, C. Z. Wang and K. M. Ho, *J. Phys. Chem. A*, 2009, **113**, 6217–6221.
- 14 X.-P. Li, W.-C. Lu, C. Z. Wang and K. M. Ho, *J. Phys.: Condens. Matter*, 2010, **22**, 465501.
- 15 B. Wang, J. Zhao, X. Chen, D. Shi and G. Wang, *Phys. Rev. A*, 2005, **71**, 033201.
- 16 G. Zhang and C. B. Musgrave, *J. Phys. Chem. A*, 2007, **111**, 1554–1561.
- 17 J. Bowlan, A. Liang and W. A. de Heer, *Phys. Rev. Lett.*, 2011, **106**, 043401.
- 18 S. Schäfer, B. Assadollahzadeh, M. Mehring, P. Schwerdtfeger and R. Schäfer, *J. Phys. Chem. A*, 2008, **112**, 12312–12319.
- 19 S. Heiles, S. Schäfer and R. Schäfer, *J. Chem. Phys.*, 2011, **135**, 034303.
- 20 D. A. Götz, S. Heiles, R. L. Johnston and R. Schäfer, *J. Chem. Phys.*, 2012, **136**, 186101.
- 21 D. A. Götz, A. Shayeghi, R. L. Johnston, P. Schwerdtfeger and R. Schäfer, *J. Chem. Phys.*, 2014, **140**, 164313.
- 22 S. Schäfer, S. Heiles, J. A. Becker and R. Schäfer, *J. Chem. Phys.*, 2008, **129**, 044304.
- 23 S. Schäfer, M. Mehring, R. Schäfer and P. Schwerdtfeger, *Phys. Rev. A*, 2007, **76**, 052515.
- 24 T. Bachels and R. Schäfer, *Rev. Sci. Instrum.*, 1998, **69**, 3794–3797.
- 25 M. Schnell, C. Herwig and J. A. Becker, *Z. Phys. Chem.*, 2003, **217**, 1003.
- 26 A. Shayeghi, D. Götz, J. B. A. Davis, R. Schäfer and R. L. Johnston, *Phys. Chem. Chem. Phys.*, 2015, **17**, 2104–2112.
- 27 P. Giannozzi, S. Baroni, M. C. N. Bonini, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. D. Corso, S. de Gironcoli, S. Fabris, G. Fratesi, R. Gebauer, U. Gerstmann, C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos, N. Marzari, F. Mauri, R. Mazzarello, S. Paolini, A. Pasquarello, L. Paulatto, C. Sbraccia, S. Scandolo, G. Sclauzero, A. P. Seitsonen, A. Smogunov, P. Umari and R. M. Wentzcovitch, *J. Phys.: Condens. Matter*, 2009, **21**, 395502.



- 28 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 29 D. Vanderbilt, *Phys. Rev. B: Condens. Matter*, 1985, **32**, 8412–8415.
- 30 A. M. Rappe, K. M. Rabe, E. Kaxiras and J. D. Joannopoulos, *Phys. Rev. B: Condens. Matter*, 1990, **41**, 1227–1230.
- 31 P. Schwerdtfeger, *ChemPhysChem*, 2011, **12**, 3143–3155.
- 32 TURBOMOLE V7.0 2015, a development of University of Karlsruhe and Forschungszentrum Karlsruhe GmbH, 1989–2007, TURBOMOLE GmbH, since 2007; available from <http://www.turbomole.com>.
- 33 M. Valiev, E. J. Bylaska, N. Govind, K. Kowalski, T. P. Straatsma, H. J. J. Van Dam, D. Wang, J. Nieplocha, E. Apra, T. L. Windus and W. de Jong, *Comput. Phys. Commun.*, 2010, **181**, 1477–1489.
- 34 F. Weigend and R. Ahlrichs, *Phys. Chem. Chem. Phys.*, 2005, **7**, 3297–3305.
- 35 B. Metz, H. Stoll and M. Dolg, *J. Chem. Phys.*, 2000, **113**, 2563–2569.
- 36 M. K. Armbruster, W. Klopper and F. Weigend, *Phys. Chem. Chem. Phys.*, 2006, **8**, 4862–4865.
- 37 M. K. Armbruster, F. Weigend, C. van Wuelen and W. Klopper, *Phys. Chem. Chem. Phys.*, 2008, **10**, 1748–1756.
- 38 M. Kühn and F. Weigend, *J. Chem. Theory Comput.*, 2013, **9**, 5341–5348.
- 39 H. Li, Y. Ji, F. Wang, S. F. Li, Q. Sun and Y. Jia, *Phys. Rev. B: Condens. Matter*, 2011, **83**, 075429.
- 40 M. K. Beyer and M. B. Knickelbein, *J. Chem. Phys.*, 2007, **126**, 104301.
- 41 D. A. Götz, S. Heiles and R. Schäfer, *Eur. Phys. J. D*, 2012, **66**, 293.
- 42 R. Antoine, M. Abd El Rahim, M. Broyer, D. Rayane and P. Dugourd, *J. Phys. Chem. A*, 2006, **110**, 10006–10011.
- 43 C. E. Moore, *Atomic Energy Levels*, Natl. Bur. Stand. Ref. Data Ser., Nat. Bur. Stand. (US) Circ. No. NSRDS-NBS 35 technical report, 1971.

